# III.A.15 Combined Theoretical and Experimental Investigation and Design of H<sub>2</sub>S Tolerant Anode for Solid Oxide Fuel Cells

## **Objectives**

- Investigate the mechanism for deterioration of solid oxide fuel cell (SOFC) anodes operating at the H<sub>2</sub>: H<sub>2</sub>S: CO ratios found in coal syngas.
- Molecular modeling of the anode materials interacting with chemical species.
- Validate the theoretical model with experimental data

## **Accomplishments**

- Established the detrimental influence of H<sub>2</sub>S on H<sub>2</sub> oxidation at anode material from quantum chemistry calculations; binding energy for H<sub>2</sub> was reduced from -80.1 kcal/mol to -21.6 kcal/mol in the presence of H<sub>2</sub>S gas molecules.
- Molecular dynamics calculations confirm that H<sub>2</sub>S gas molecules slow the diffusion of H<sub>2</sub> molecules towards anode material.
- Analyzed the impact of CO on the mixture of H<sub>2</sub> and H<sub>2</sub>S; CO decreased the diffusion coefficient of H<sub>2</sub>S.

#### Introduction

Coal syngas is the gas produced from the burning of coal and has a good amount of hydrogen along with other chemical species as such as CO and  $\rm CO_2$ . However, coal syngas also contains  $\rm H_2S$  at high concentrations between 0.5-5% depending on where the coal is mined [1]. Of the different types of fuel cell, the solid oxide fuel cell (SOFC) is the most viable option to use coal syngas as its fuel supply because it operates at 850-1000°C. It is also known that SOFCs can handle

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CO and CO<sub>2</sub> as contaminants in the H<sub>2</sub> supply because of its high operating temperature. But the presence of H<sub>2</sub>S in the fuel stream will deteriorate the anode material. Most of the research has been focused on developing new anode materials for SOFCs rather than studying the cause of the anode deterioration due to H<sub>2</sub>S. In this project, we propose to use a systematic approach to understanding the effect of H<sub>2</sub>S from syngas on the anode materials (Ni-YSZ, Ni-CeO<sub>2</sub>-YSZ) with molecular modeling computations, and to perform experiments to validate the models.

## **Approach**

In this study, molecular modeling is used to understand the interaction of  $\rm H_2S$  with anode materials (Ni-YSZ, Ni-CeO<sub>2</sub>-YSZ). Molecular modeling consists of first principle quantum chemistry (QC) calculations using Gaussian 03 software to optimize the structure of each compound and the systems consisting of anode and different chemical species. Physical properties, such as diffusion coefficient of the chemical species and their interactions with anode material, are calculated using molecular dynamics (MD) simulations with Cerius2 (v 4.8) software. Finally, an experimental set-up will be constructed to perform SOFC experiments with different anode materials and validate the mathematical models.

#### **Results**

Molecular modeling of different SOFC anode material chemical species was studied. First, the electrolyte of SOFC, yttria-stabilized zirconia (YSZ), was optimized by QC calculations using Gaussian 03 software. NiO was added to the optimized YSZ structure to form Ni-YSZ, the anode material. All the possible systems consisting of Ni-YSZ (anode), H<sub>2</sub>, H<sub>2</sub>S and CO were constructed and quantum calculations were performed. The binding energies for each system are shown in Table 1. The binding energy for each system has a negative value which means the oxidation of the components on the Ni-YSZ surface is thermodynamically feasible. In case of single molecules, oxidation of H<sub>2</sub> is favored more than oxidation of H<sub>2</sub>S and CO because the binding energy for H<sub>2</sub> (-80.1 kcal/mol) is lower than H<sub>2</sub>S (-21.4 kcal/mol) and CO (-23.9 kcal/mol).

The binding energy for the system where  $H_2S$  is combined with  $H_2$  (-21.6 kcal/mol) is higher than the system with only  $H_2$ , which means  $H_2$  oxidation on the Ni-YSZ surface is obstructed by the presence of  $H_2S$  molecules. But when CO is mixed with  $H_2$ , the binding

**TABLE 1.** Binding Energies for Various Combinations of Gas Components

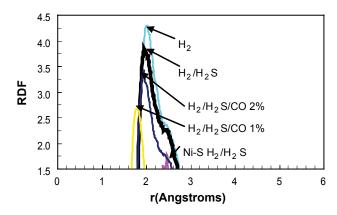
Systems	Binding Energy (kcal/mol)		
H <sub>2</sub>	-80.1		
H <sub>2</sub> S	-21.4		
CO	-23.9		
$H_2 + H_2S$	-21.6		
H <sub>2</sub> + CO	-155.5		
H <sub>2</sub> S + CO	-134.6		

energy (-155.5 kcal/mol) is far less than the binding energy for the  $\rm CO+H_2S$  system (-134.6 kcal/mol), which suggests that oxidation of both CO and  $\rm H_2$  on the surface of Ni-YSZ anode is more favored than the  $\rm CO+H_2S$  system.

The computational chemistry model was extended and MD calculations were performed to understand the interactions between the atoms in the anode and the chemical species (H<sub>2</sub>, H<sub>2</sub>S and CO) in its environment. The radial distribution function (RDF) was used to determine the packing and orientation of the chemical species and anode material. The RDF plot gives the interaction of each chemical species with different atoms in the anode as a function of distance from the surface of the anode.

One of the RDF plots for the H<sub>2</sub> system at 25°C showed that H<sub>2</sub> has a higher affinity for the Ni atom in the anode than other atoms (Y, Zr and O), which means H<sub>2</sub> oxidizes at Ni sites in the Ni-YSZ anode. Figure 1 shows the RDF plot for the Ni-H<sub>2</sub> pair at 850°C in the presence of H<sub>2</sub>S and CO molecules. The pure H<sub>2</sub> system has the highest peak value, which decreases with the addition of H<sub>2</sub>S and CO to the mixture. This proves again that oxidation of H<sub>2</sub> at Ni sites are slowed down by the presence of H<sub>2</sub>S and CO molecules. In addition, one can observe the position of the Ni-S pair from the H<sub>2</sub>/H<sub>2</sub>S system, which is farther away from the anode surface as compared to the Ni-H<sub>2</sub> pair. This confirms that H<sub>2</sub> is more likely to get oxidized at Ni sites than the H<sub>2</sub>S molecule. Interaction of H<sub>2</sub>S with different atoms in the anode and H<sub>2</sub> molecules at 850°C is shown in Figure 2. The RDF plot shows that S forms a pair with Ni and H<sub>2</sub> rather than with any other atoms. The Ni-S pair is slightly farther away from the anode than the Ni-H<sub>2</sub> pair.

Other than the RDF plot, MD calculations can be used to find the diffusion coefficient of each gas molecule. In Table 2 the diffusion coefficients for  $\rm H_2$ ,  $\rm H_2S$  and CO in various systems are listed. Diffusivity for pure  $\rm H_2$  was found to be 1.35 x  $\rm 10^{-4}$  cm²/s, but when  $\rm H_2S$ 



**FIGURE 1.** RDF Plot for Ni-H $_2$  Pair at 850°C with the Presence of H $_2$ S and CO

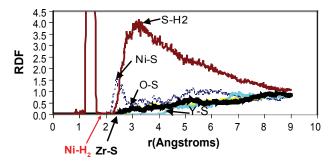


FIGURE 2. Interaction of  $\rm H_2S$  Molecule with Ni, Y, 0, Zr Atoms and  $\rm H_2$  Molecule at  $850^\circ$ 

gas was mixed with  $H_2$ , the diffusion coefficient of  $H_2$  was lowered by 20% to 1 x  $10^{-4}$  cm²/s. This observation agrees with QC calculations that in a  $H_2+H_2S$  mixture, oxidation of  $H_2$  takes place at a slower pace due to transport limitations. Now with the addition of CO to  $H_2$  gas, the  $H_2$  diffusivity has been dropped to 5 x  $10^{-6}$  cm²/s, which is 96% lower than the pure  $H_2$  system. In the case of a  $H_2+H_2S(2\%)+CO(2\%)$  mixture, CO slows down both the diffusion of  $H_2$  (9 x  $10^{-6}$  cm²/s) and also  $H_2S$  (2 x  $10^{-9}$  cm²/s).

**TABLE 2.** Diffusivities of the Gas Components at  $850^{\circ}$  C for Different Systems

Systems	H <sub>2</sub> Diffusivity (cm²/s)	H <sub>2</sub> S Diffusivity (cm²/s)	CO Diffusivity (cm²/s)
H <sub>2</sub>	1.35 x 10 <sup>-4</sup>	_	_
$H_2 + H_2 S (2\%)$	1 x 10 <sup>-4</sup>	1 x 10 <sup>-5</sup>	-
H <sub>2</sub> + CO (2%)	5 x 10 <sup>-6</sup>	-	3 x 10 <sup>-7</sup>
$H_2 + H_2S (1\%) + CO (1\%)$	2 x 10 <sup>-5</sup>	8 x 10 <sup>-9</sup>	6 x 10 <sup>-9</sup>
$H_2 + H_2S (2\%) + CO (2\%)$	9 x 10 <sup>-6</sup>	2 x 10 <sup>-9</sup>	6 x 10 <sup>-9</sup>

## **Conclusions and Future Directions**

- Quantum chemistry calculation proves oxidation of pure H<sub>2</sub> is more thermodynamically favored on the Ni-YSZ surface than pure H<sub>2</sub>S and CO.
- RDF plots for H<sub>2</sub> indicate that H<sub>2</sub> has a greater affinity for the Ni atom than any other atoms in the anode material.
- Diffusion of H<sub>2</sub> was slowed down in the presence of H<sub>2</sub>S and CO, but when all the three gases are mixed, CO reduces the diffusion coefficient of H<sub>2</sub>S.
- In the future, we would like to include N<sub>2</sub> and H<sub>2</sub>O (moisture) atoms to the anode environment and perform QC and MD calculations.
- Perform molecular modeling with another anode material – Ni-CeO<sub>2</sub>-YSZ.
- Design and construct a SOFC experimental set-up to verify the theoretical model.

#### **FY 2006 Publications/Presentations**

- 1. "Theoretical Investigation of NiYSZ in the Presence of  $H_2S$ " Marquez, A., De Abreu, Y., and Botte, G. G., *Electrochemical and Solid-State Letters*, 9 (3) A163-A166 (2006).
- **2.** "Theoretical Investigations of Solid Oxide Fuel Cell Anode Materials" AIChE 2005 Annual Meeting, October 30<sup>th</sup> November 4<sup>th</sup>, 2005, Cincinnati, OH.
- 3. "Theoretical Investigations of Solid Oxide Fuel Cell Anode Materials in the Presence of  $\rm H_2/H_2S/CO$ " The  $\rm 31^{st}$  International Technical Conference on Coal Utilization & Fuel Systems, The Clearwater Coal Conference, May 21-25, 2006, Clearwater, FL.

#### References

1. G. Y. Lai, *High Temperature Corrosion of Engineering Alloys*, ASM International, Materials Park, OH (1990) p. 117.